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# Closure Effect of I + II Mixed-mode Crack for EA4T Axle Steel



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# Abstract

The crack-closure effect is a crucial factor that affects the crack growth rate and should be considered in simulation analysis and testing. A mixed-mode I + II loading fatigue crack growth test was performed using EA4T axle steel specimens. The variation of the plastic-induced crack closure (PICC) effect and the roughness-induced crack closure (RICC) effect during crack deflection in the mixed-mode is examined in this study. The results show that the load perpendicular to the crack propagation direction hinders the slip effect caused by the load parallel to the crack propagation direction under mixed-mode loading, and the crack deflection is an intuitive manifestation of the interaction between the PICC and RICC. The proportion of the  $R_A$  value change on the crack side caused by contact friction was reduced by the interaction between PICC and RICC. The roughness of the crack surface before and after the crack deflection is different, and the spatial torsion crack surface is formed during the crack propagation process. With the increase of the crack length, the roughness of the fracture surface increases. During the crack deflection process, the PICC value fluctuates around 0.2, and the RICC value is increased to 0.15.

**Keywords** Crack closure, Crack deflection, Plasticity-induced closure, Roughness-induced closure, Interaction mechanism

# 1 Introduction

Fatigue damage and fracture are inevitable phenomena in the service life of engineering structures and materials [1]. When fatigue damage accumulates to a certain level, local plastic deformation occurs at the location of the highest stress region. Plastic deformation is one of the significant reasons for the occurrence of crack closure effects, and the existence of crack closure effects is conducive to restraining the opening of the crack. The crack closure effect encompasses factors such as plasticity, roughness, the presence of oxides, the small crack phenomenon, overload, and compression overload [2–4]. A reasonable understanding and application of the closure effect contribute to researchers in engineering structure design and lifespan assessment.

Cracks can be divided into three types based on linear elastic fracture mechanics (LEFM): opening (type I), sliding (type II), and tearing types (type III), the different operating conditions in actual service lead to fatigue cracks in either isolated or mixed forms. To provide a reasonable description or assessment of the process of crack propagation, researchers have conducted long-term and systematic theoretical and experimental studies [5-9], including methods related to the Elber crack closure model, the incremental (or partial) crack closure model, and the zero crack propagation load method. With the help of relevant experimental methods, the relationship between the crack growth rate determined by far-field loading conditions, crack geometry and the stress intensity factor range ( $\Delta K$ ) under small-scale yield conditions is established. The contribution of oxidation or roughness to crack closure is shown to be smaller than the contribution of the slope change of the load-displacement



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curve. In addition, the plastic strain-related mechanism at the crack tip can explain most of the test results, but some test results cannot be explained according to the plastic-induced crack closure (PICC) effect, and even contradict the crack closure method, other factors may play a role, such as the degradation of material mechanical properties (hardening and softening) during creepfatigue process, and interaction between micro defects and microstructure (metal inclusions) [10, 11].

Research on the crack closure effect mainly focuses on PICC, roughness-induced crack closure (RICC) and oxide-induced crack closure (OICC) effects. Under the condition of uniaxial loading, materials are prone to PICC. Under mixed-mode loading conditions, crack deflection leads to mutual contact between fracture surfaces, resulting in the RICC effect. In humid environments, some materials are subject to OICC effects, as fatigue crack surfaces come into contact with the air, initiating oxidation-reduction reactions that accumulate oxidation-reduction products on the fracture surfaces. Peters and Yamaguchi et al. [12, 13] presented a non-contact full-field optical measurement method to investigate the PICC effects. McClung [14] used an elastic-plastic finite element method (FEM), which is considered PICC, to simulate fatigue crack propagation (FCG), the traditional Irwin-Rice estimate for crack tip plastic zone size in plane stress is found to be generally consistent with the FEM results. Christopher et al. [15, 16] established a new crack-tip stress field model (Christopher-James-Patterson, CJP model) based on the traditional Westergaard's model, the results show that the CJP model can better explain the PICC effect. Yang et al. [17] used the Levenberg-Marguardt (L-M) iterative method to solve for the crack tip position and apply it to the CJP crack tip field model. Li et al. [18] considered the effect of the dislocation field on the plastic zone of the crack tip and hence on the elastic field by introducing a plastic flow factor  $\rho$ , the plastic flow correction provides a more accurate crack tip field model and improves the CJP crack growth relationship. Yang et al. [19] applied the CJP model to the FCG of U71MnG rail steel standard compact tensile (CT) specimens, the results point out that the CJP model can describe the contour of the plastic zone, and can characterize the influence of crack tip compatible stress and the crack closure effect on crack propagation under different stress ratios. Paysan et al. [20] applied an elastic-plastic 3D FEM to study the evolution of plasticity during FCG. The result can reproduce the major features of fatigue crack closure, the 3D and load-dependent evolution of PICC. Gonzales et al. [21] provided an experimental methodology to determine the reversed plastic zone size at the crack tip under cyclic loading through high-resolution digital image correlation (DIC) measurements.

Although modern X-ray microtomography 3D technology can achieve sub-micron resolution, cracks cannot be regarded as simple planar discontinuities in isotropic media, there is no direct method to measure the actual development of full-thickness crack closure, so the FEM is generally used in the study of RICC or OICC. Chen et al. [22] introduced elastic wedges into elastic cracks to simulate RICC or OICC effects and applied LEFM to solve the relationship between externally applied stress and crack opening displacement. Field et al. [23] conducted a general fractographic analysis of the fatigue fracture surface to understand what correlation there may be between surface roughness and the FCG rate. Gonzalez et al. [24] reviewed and discussed the key points on the RICC that are considered to affect crack tip shielding, the dislocation-based models have indicated that the crack tip shielding effect from a single asperity is relatively minor. Pippan et al. [25] explored the RICC effect from asymmetric dislocations and measured the local crystal orientation of fatigue cracks in the near-threshold region of Cu and Fe. Crapps et al. [26] studied the crack-closure problem under the combined action of PICC and RICC by introducing a serrated crack wake and changing the roughness and geometric shape period of the crack surface. The results show that the average crack closure degree increased with an increase in the roughness of the crack surface, whereas the geometric shape period has little effect on RICC. Pokluda et al. [27, 28] characterized the size of the crackclosure force component caused by roughness, the results are superior to the CJP model in evaluating the closure force component caused by roughness.

The FCG behavior in the mixed-mode has been further investigated since the introduction of compact tensile shear (CTS) specimens and relative loading devices by Richard [29]. Previous studies have reported that the effect of the  $L/r_p$  ratio (L and  $r_p$  represent the crack roughness length scale and plastic zone size, respectively) during a single peak overload may be similar to the RICC under constant amplitude (CA) loading [30, 31]. Parry et al. [32] studied the opening behavior of the crack tip section and the residual shear deformation of the asperity in the crack wake and described the crack deflection under CA loading from the perspective of micromechanics. Singh et al. [33] discussed and analyzed the competitive relationship between RICC and the PICC process after an overload. The material exhibited different degrees of deflection in mixed-mode loading conditions [34]. Under pure mode-I loading, FCG is not affected by RICC. However, under mixed-mode loading, it is affected by the mutual contact and interaction forces of the fracture surface. Previous work [35] showed that the crack propagation behavior

can be described more effectively because the solution includes PICC and RICC effects, compared with the traditional mixed model crack growth models with higher fitting correlation coefficients. The relevant studies of OICC are not described in this paper, because no redox reaction was found in the previous experiments, and no oxide accumulation was observed on the fracture surface.

The above studies are mainly devoted to exploring or explaining the mechanism of the PICC or RICC effect while proposing new means or using FEM and advanced technologies et al. to simulate the crack propagation process, and most of the studies focus on uniaxial loading conditions. Few literatures have described the interaction between PICC and RICC during crack deflection in the mixed-mode so far, so it is necessary to further study the crack closure effect in the process of crack deflection. In this study, EA4T axle steel was used to explore the variation of PICC and RICC under mixed-mode I + II loading conditions and the interaction between the two crack closure effects during crack deflection propagation. The DIC technique and the replica method were employed to record the crack tip displacement field, deflection angle and crack length, the change of SIFs during crack deflection was simulated by FEM.

# 2 Calculation of the Crack Closure

## 2.1 PICC Calculation

The CJP model established by Christopher et al. [36] considers the influence of compressive stress between crack surfaces and shear stress between crack sides, as shown in Figure 1, it shows the idealized diagram of the elastic-plastic interface stress at the crack tip during fatigue crack propagation ( $F_A$  is mode–I force;  $F_B$  is an additional force caused by mode–II loads;  $F_{\rm T}$  is a force caused by positive T-stress;  $F_{\rm S}$  is interface shear force in the elastic–plastic interval;  $F_{\rm C}$  and  $F_{\rm P}$  produce a shielding effect together;  $F_{\rm C}$  is the contact force between the sides of the crack, and  $F_{\rm P}$  is the force generated by the compatibility constraint of the plastically deformed material.). The crack tip plastic zone gradually transitions from the crack front to the crack wake with the crack propagation, a plastic inclusion is formed at the crack tip and a plastic wedge is formed between the crack sides to cause the PICC effect. The two effective crack propagation driving factors are used in the model, as shown in Eq. (1).

$$\begin{cases} \Delta K_{\rm cjp} = (K_{\rm F,max} - K_{\rm R,max}) - (K_{\rm F,min} - K_{\rm R,min}), \\ \Delta K_F = K_{\rm F,max} - K_{\rm F,min}, \end{cases}$$
(1)

where  $K_{\rm F}$  is the SIF that drives crack growth, and  $K_{\rm R}$  is the SIF that acts to retard crack growth.  $K_{\rm F,max}$ ,  $K_{\rm R,max}$ ,  $K_{\rm F,min}$ ,  $K_{\rm R,min}$  at  $K_{\rm F}$  and  $K_{\rm R}$  under the maximum and minimum stress states.

The CJP model is defined as [36]:

$$\begin{aligned} \left|\sigma_{y} - \sigma_{x} + 2i\sigma_{xy}\right| &= (A_{r} + i3B_{i})z^{-\frac{1}{2}} + (B_{r} + iB_{i})z^{-\frac{3}{2}}\overline{z} \\ &+ Cz^{0} + Dz^{-\frac{1}{2}}\ln(z) + Ez^{-\frac{3}{2}}\overline{z}\ln(z). \end{aligned}$$
(2)

The elastic stress fields near the crack tip are given by:

$$\begin{aligned} \sigma_{y} &= \frac{1}{2} (A_{r} - 4B_{r} - 8E)r^{-\frac{1}{2}} \cos \frac{\theta}{2} + \frac{1}{2}B_{r}r^{-\frac{1}{2}} \cos \frac{5\theta}{2} + \\ &\quad \frac{1}{2}B_{l}r^{-\frac{1}{2}} \left( \sin \frac{5\theta}{2} - \sin \frac{\theta}{2} \right) + \frac{1}{2}Er^{-\frac{1}{2}} \times \\ &\quad \left[ \ln (r) \left( \cos \frac{5\theta}{2} - 5 \cos \frac{\theta}{2} \right) + \theta \left( \sin \frac{5\theta}{2} - 5 \sin \frac{\theta}{2} \right) \right] + o \left( r^{\frac{1}{2}} \right), \\ \sigma_{x} &= -\frac{1}{2} (A_{r} + 4B_{r} + 8E)r^{-\frac{1}{2}} \cos \frac{\theta}{2} - \frac{1}{2}B_{r}r^{-\frac{1}{2}} \cos \frac{5\theta}{2} - \\ &\quad \frac{1}{2}B_{l}r^{-\frac{1}{2}} \left( \sin \frac{5\theta}{2} + 7 \sin \frac{\theta}{2} \right) - \frac{1}{2}Er^{-\frac{1}{2}} \times \\ &\quad \left[ \ln (r) \left( \cos \frac{5\theta}{2} + 3 \cos \frac{\theta}{2} \right) + \theta \left( \sin \frac{5\theta}{2} + 3 \sin \frac{\theta}{2} \right) \right] - \\ &\quad C + o \left( r^{\frac{1}{2}} \right), \\ \sigma_{xy} &= -\frac{1}{2}r^{-\frac{1}{2}} \left( A_{r} \sin \frac{\theta}{2} + B_{r} \sin \frac{5\theta}{2} \right) + \\ &\quad \frac{1}{2}B_{l}r^{-\frac{1}{2}} \left( \cos \frac{5\theta}{2} + 3 \cos \frac{\theta}{2} \right) - \\ &\quad Er^{-\frac{1}{2}} \sin \theta \left[ \ln (r) \cos \frac{3\theta}{2} + \theta \sin \frac{3\theta}{2} \right] + o \left( r^{\frac{1}{2}} \right). \end{aligned}$$
(3)

The  $K_{\rm F}$  and  $K_{\rm R}$  can be obtained in Eq. (4), which represent the driving SIF and the delaying SIF respectively.

$$\begin{cases} K_F = \lim_{r \to 0} \left[ \sqrt{2\pi r} \left( \sigma_y + 2Er^{-\frac{1}{2}} \ln \left( r \right) \right) \right] \\ = \sqrt{\frac{\pi}{2}} (A_r - 3B_r - 8E), \\ K_R = \lim_{r \to 0} \left( \sqrt{2\pi r} \sigma_x \right) = -4\sqrt{\frac{\pi}{2}} (2B_i + E\pi). \end{cases}$$
(4)

The five-parameter can be solved from displacement fields:

$$\begin{split} & 2\mu \left( u_{x} + iu_{y} \right) \\ & = \kappa \left( -2(B_{r} + iB_{i})z^{\frac{1}{2}} - 2Ez^{\frac{1}{2}}\ln\left(z\right) - \frac{C}{4}z \right) - \\ & z \left( -(B_{r} + iB_{i} + 2E)\overline{z}^{-\frac{1}{2}} - E\overline{z}^{-\frac{1}{2}}\overline{\ln\left(z\right)} - \frac{C}{4} \right) - \\ & \left[ (A_{r} + i3B_{i})\overline{z}^{\frac{1}{2}} + D\overline{z}^{\frac{1}{2}}\overline{\ln\left(z\right)} - 2D\overline{z}^{\frac{1}{2}} + \frac{C}{2}\overline{z} \right], \end{split}$$
(5)

where  $u_x$  and  $u_y$  are the horizontal and vertical displacements respectively,  $\mu = E/2(1+\nu)$ ;  $k = 3-4\nu$  (plane



Figure 1 Idealised diagram of crack tip elastic-plastic interface force

strain) or  $k = (3-\nu)/(1+\nu)$  (plane stress);  $u_x$  and  $u_y$  are shown explicitly below with the assumption D+E = 0. After obtaining the crack tip displacement field, the equivalent stress value is solved according to the generalized Hooke's law. The value is compared with the yield strength value of the material to distinguish which stress values of the pixels obtained by DIC technology are in the linear elastic range. The position information of this part of the points is stored in the document. Substituting the displacement values of all points in the document into Eqs. (6) and (7), a linear equation set can be formed. The linear fitting method can be used to solve the parameters in the CJP model, and then the parameters such as  $K_F$  and  $K_R$  can be obtained.

$$2\mu u_{x} = r^{\frac{1}{2}} (-A_{r} - 2B_{r}\kappa - 2E) \cos\frac{\theta}{2} + r^{\frac{1}{2}} (2B_{i}\kappa - 3B_{i}) \sin\frac{\theta}{2} + r^{\frac{1}{2}} (B_{r} + 2E) \cos\frac{3\theta}{2} - r^{\frac{1}{2}} B_{i} \sin\frac{3\theta}{2} - \frac{C}{4} r(1 + \kappa) \cos\theta + r^{\frac{1}{2}} E \ln(r) \left(\cos\frac{3\theta}{2} + (1 - 2\kappa) \cos\frac{\theta}{2}\right) + r^{\frac{1}{2}} E \theta \left(\sin\frac{3\theta}{2} + (1 + 2\kappa) \sin\frac{\theta}{2}\right),$$
(6)

$$2\mu u_{y} = r^{\frac{1}{2}} (-2B_{i}\kappa - 3B_{i}) \cos \frac{\theta}{2} + r^{\frac{1}{2}} (A_{r} - 2B_{r}\kappa + 2E) \sin \frac{\theta}{2} + r^{\frac{1}{2}} (B_{r} + 2E) \sin \frac{3\theta}{2} + r^{\frac{1}{2}} B_{i} \cos \frac{3\theta}{2} + r^{\frac{1}{2}} E \ln (r) \left( \sin \frac{3\theta}{2} - (1 + 2\kappa) \sin \frac{\theta}{2} \right) - r^{\frac{1}{2}} E \theta \left( \cos \frac{3\theta}{2} + (1 + 2\kappa) \cos \frac{\theta}{2} \right) + \frac{C}{4} r(3 - \kappa) \sin \theta.$$
(7)

The crack closure ratio proposed by Pokluda [37] is

$$\frac{K_{PICC}}{K_{\max}} \approx 2c, \tag{8}$$

where  $K_{\text{PICC}}$  is the SIF value calculated after considering PICC;  $K_{\text{max}}$  is the maximum SIF value;  $c=b/10(\pi)^{0.5}B$ ; *b* is the distance between the upper and lower grains of the material,  $B \approx 10^{-10}$  m; *c* is a dimensionless constant independent of the material,  $c \approx 0.1$ . It should be noted that the evaluation of PICC in the Pokluda model is relatively straightforward, requiring only certain material parameters for calculation. The listing of this expression is for analysis, comparison, and validation of the results obtained in this experiment.

In related studies, the load corresponding to crack opening is denoted as  $P_{op}$ , and U is used to represent the degree of crack closure [38, 39], as shown in Eq. (9). In this study, to quantify the PICC component, the  $P_{op}/P_{max}$  value is employed to represent the crack closure component. Based on the force-displacement data recorded during the experiment, combined with the parabolic-linear method [40], the crack opening force at each stage of crack deflection can be determined.

$$\mathcal{U} = \frac{P_{\max} - P_{op}}{P_{\max} - P_{\min}},\tag{9}$$

where *U* is a dimensionless constant;  $P_{\text{max}}$  and  $P_{\text{min}}$  are the maximum and minimum load, respectively.

## 2.2 RICC Calculation

The asymmetry of dislocations and roughness of the fracture surface led to different shear displacements on both sides of the crack, resulting in the RICC effect. Assuming that the roughness geometry is simple, as shown in Figure 2, the dotted line indicates the crack propagation direction. The calculation expression is given in Eq. (10) [27].

$$\frac{K_{RICC}}{K_{\max}} = C\eta \sqrt{\left(R_A^2 - 1\right)} + \frac{3\eta(R_A - 1)}{2\sqrt{6} + 6(R_A - 1)}$$
(10)

where  $K_{\text{RICC}}$  is the RICC SIF;  $K_{\text{max}}$  is the maximum SIF, and  $R_{\text{A}}$  is the roughness of the fracture.  $\eta$  is a statistical parameter, it obeys the Weibull distribution.  $S_{\text{Rc}} \in (0.2,1)$ is a fitting parameter, and a value of 0.5 is generally acceptable [41].  $d_{\text{m}}$  is the characteristic microstructure distance, and  $r_{\text{p}}$  is the static plastic zone size [28].

A theoretical analytical solution of the CJP model in mixed-mode is presented in the previous work, as shown in Eq. (12). The crack deflection propagation behavior can be described more effectively by considering the influence of PICC and RICC [35]. The SIF value in the process of crack deflection is calculated by Eq. (12) in this study.

$$\begin{cases} \Delta K_{P-R} = x \left( K_{F,\max} - K_{R,\max} \right) - \left( K_{F,\min} - K_{R,\min} \right), \\ x = 1 - C\eta \sqrt{R_A^2 - 1} - \frac{3\eta(R_A - 1)}{2\sqrt{6} + 6(R_A - 1)}, \end{cases}$$
(12)

where  $\Delta K_{\text{P-R}}$  is the  $\Delta K_{\text{eff}}$  value with the consideration of PICC and RICC effects, and *x* is the linear scaling factor.  $R_{\text{A}}$  value can be obtained by measuring the fracture area after the testing.

## **3 Experimental Procedure**

The sample material used in this study was EA4T axle steel. The specimens were subjected to a mixed-mode I + II loading FCG test, which was performed using an electronic tensile-torsion testing machine (INSTRON E10BMT). For the formal test,  $P_{max} = 4$  kN, R = 0.1, and f = 25 Hz, and CA loading was carried out in the directions of 30°, 45°, and 60°. During the test, a replica method was adopted to record the crack propagation trajectory, so the crack tip location and angle can be defined after a certain number of cycles. The crack propagation length was measured after a specific number of cycles, and loaded at a low speed (the maximum load remained unchanged). The sampling frequency was adjusted to



**Figure 2** Diagram of maximum long-range RICC induced by asymmetric dislocation array

1000 Hz to collect sufficient load-displacement data. A DIC device, i.e., Revealer 2D-DIC was used to obtain the displacement field of the FCG and calculate the change in the SIF value at the crack tip during crack deflection. The resolution of DIC equipment is 0.00638 pixels/mm. An industrial microscope with a resolution of 10  $\mu$ m was used to observe and record crack propagation paths in real-time.

The fatigue crack propagation process in the mixedmode can be divided into two stages. In the first stage, under the interaction of shear and tensile stresses perpendicular to the prefabricated crack surface, cracks initiated and propagated in the surface slip band under shear stress. In the second stage, crack propagation was approximately perpendicular to the principal tensile stress, as shown in Figure 3a. As the crack propagated from the prefabricated crack position to the first stage of crack propagation, the slip system formed a certain angle with the crack propagation direction, resulting in crack deflection. The mixedmode loading device and crack deflection are shown in Figure 3a, the fixture designed by Richard was employed for loading [29]. The dimensions of the specimen are shown in Figure 3b, and the field test device is shown in Figure 4. The chemical compositions and basic mechanical properties of the materials are listed in Tables 1 and 2, respectively, where Et. is element, Wt. is weight (%). *E* is elastic modulus (GPa),  $\nu$  is poisson's ratio,  $\sigma_s$  is yield strength,  $\sigma_{\rm b}$  is tensile strength (MPa),  $\delta$  is elongation (%),  $\Psi$ is reduction of area (%) [42]. It is worth noting that the degradation phenomenon of material mechanical properties is not considered in this study, mechanical properties, such as yield strength, ultimate tensile strength and elongation are key factors in damage evaluation. Sun et al's work showed that the degradation of mechanical properties is influenced by cyclic softening/hardening characteristics of materials, and a comprehensive characterization method is proposed using a tensile plastic strain energy density parameter [43].

## **4** Results and Discussions

## 4.1 Simulation Results

The research shows that the FEM is used to simulate the crack propagation process, which can reflect the change trend and range of each parameter to a certain extent. In Singh et al's work [33], a single crack deflection angle of  $\theta$  = 45° and deflection lengths between L = 18.75 and L = 225 µm are considered, fatigue surface roughness is idealized as a simple regular zigzag geometry for RICC studies. This modelling method is convenient for researchers to explore the changing trend of the RICC effect, but the crack deflection angle is fixed, while the crack deflection angle of the material or structure in service is changed in real time. In Nittur et al's work [44], only the right half of the specimen is modelled due to symmetries of the loading and geometry



Figure 3 Crack deflection and test loading diagram: a crack deflection stage and fixture, b geometric dimensions of CTS specimen



Figure 4 Test device (loading angle  $\alpha$ =60°)

and appropriate boundary conditions are imposed along the vertical symmetry line. The model consists of 16 layers of 8 noded isoparametric 'C3D8R' brick elements and has finer mesh towards the free surface and coarser mesh towards the mid-plane. The crack propagation is accomplished by releasing the constraints on the current crack front nodes at the end of the load cycle. Since the crack deflection occurs during the mixed-mode crack propagation process, the submodel technique was not used in this study. In Antunes et al's work [45], the finite element mesh is refined at the crack front to model the severe plastic deformation gradients and enlarged at remote positions to reduce the numerical effort. In this study, the model crack propagation path element was also preprocessed to reasonably simulate the crack propagation process.

In this study, a commercial finite element software Franc3D 8.0 was used to simulate FCG in mixed-mode. The element type was selected as an 8-node isoparametric plane stress element (CPS8 element). According to Richard distribution equation [46], the CTS specimens were subjected to force and constraint. It is worth noting that in the study of elastic-plastic stress (strain), simulation results are influenced by the loading path when plastic stress/deformation is involved. For instance, if tensile loading is applied before shear loading or shear loading is applied before tensile loading, the results can differ significantly [47, 48]. The above problems are avoided by using the Richard distribution equation for loading and constraint setting.

CTS modelling was established in the finite element software of HyperWorks 21.0. The cell grid was divided into three parts based on its size: the first part was a circular area with a diameter of 14 mm centred on the crack tip. Based on the sensitivity analysis results of Wang et al. [34], the cell grid size of this part was 0.05 mm. The grid size increases from the inside to the outside, and the other two parts of the grid size are 0.1 mm and 0.2 mm, respectively, as shown in Figure 5. Material parameters were set according to the values provided in Table 2, and Paris constants were determined based on experimental fitting results, with  $c = 4.92 \times 10^{-9}$  and m = 2.1676 [35]. A 2 mm elliptical crack at the initial crack tip throughout the entire thickness of the CTS

Table 1 Chemical composition of EA4T (%)

Et.	c	Si	Mn	Р	S	Cr	Cu	Ni	Мо
Wt.	0.27	0.39	0.72	0.0075	0.0013	1.11	0.014	0.25	0.247

Table 2 Mechanical properties of EA4T

Ε	v	σ	σ <sub>b</sub>	δ	Ψ
209	0.3	520	695	20.88	63.76

specimen was inserted. Subsequently, a combination of the extended finite element method (XFEM) was used to simulate mixed-mode crack propagation. This study calculated the SIF based on the displacement extrapolation method [49], as shown in Eq. (13).

$$\begin{cases} K_{\rm I} = \lim_{r \to \infty} 0.25 E (2\pi/r)^{0.5} u_2, \\ K_{\rm II} = \lim_{r \to \infty} 0.25 E (2\pi/r)^{0.5} u_1, \end{cases}$$
(13)

where *E* is Young's modulus, *r* is the distance from the crack tip,  $u_1$  and  $u_2$  are the opening displacement and sliding displacement of the two relative points above and below the crack.

The maximum tangential stress (MTS) [50] and strain energy density factor (*S*) criteria [51] were sat as the mixed-mode crack growth criterion in this paper, the theoretical expressions of crack propagation direction  $\theta$ of MTS criterion and *S* criterion are shown as follows:

MTS criterion:

$$\theta = \arccos \frac{3K_{II}^2 + \sqrt{K_I^4 + 8K_I^2 K_{II}^2}}{K_I^2 + 9K_{II}^2},$$
(14)

S criterion:

$$\frac{\partial S}{\partial \theta} = 0 \text{ and } \frac{\partial^2 S}{\partial^2 \theta} > 0,$$
 (15)

where *S* is the strain energy density coefficient,  $S = a_{11}K_1^2 + 2a_{12}K_1K_{II} + a_{22}K_{II}^2$ , the coefficients  $a_{11} = (1+\nu/8\pi E)$  $(1+\cos\theta)(k-\cos\theta)$ ,  $a_{12} = (1+\nu/8\pi E)\sin\theta[2\cos\theta-(k-1)]$ ,  $a_{22} = (1+\nu/8\pi E)[(k+1)(1-\cos\theta)+(1+\cos\theta)(3\cos\theta-1)]$ . *E* and *v* are Young's modulus and Poisson's ratio, respectively,  $k = (3-\nu)/(1+\nu)$  under plane stress condition and  $k = 3 - 4\nu$  under plane strain condition.

As there was a small difference between the calculated values of the two criteria in previous work, only the MTS results were plotted here, as shown in Figure 6. The variation in  $K_{I}$  and  $K_{II}$  values with crack length is depicted in the figure,  $K_{\rm I}$  and  $K_{\rm II}$  are the SIF values of mode-I and mode-II loading conditions, respectively. In Figure 6a, with increased crack propagation length, the  $K_{I}$  value shows an overall upward trend, consistent with the results obtained using the Westergaard method. The curve in Figure 6a showed an alternating trend of increase and decrease during ascent, which was related to crack deflection. The a-b segment in Figure 6b is defined as the first stage and the b-c segment is defined as the second stage, the dotted line is the fitted line of the simulation data, which has no clear physical meaning for the following analysis. The  $K_{II}$  value gradually increased with increasing crack length in the first stage and decreased in the second stage. The growth trend of the  $K_{II}$  value in the first stage was similar to that of the Paris crack propagation curve. As the crack length increased, the contact area of the crack side also increased. In combination with the crack deflection diagram in Figure 3a, the variation



**Figure 5** Exemplary FEA results of mixed-mode loading ( $\alpha = 45^{\circ}$ )

trend of  $K_{\rm II}$  value is explained. As shown in Figure 7,  $P_{\rm I}$ and  $P_{\rm II}$  are the loads corresponding to the directions perpendicular and parallel to the crack propagation, the a-c in Figure 7 shows the load projection when the mixed mode's loading angle is set to 30 °, 45 ° and 60 ° respectively. The first stage region in Figure 3a was non-contact in the mixed-mode loading when the crack propagated, and as the crack deflected (the stages 1 to 2), although the  $P_{\rm II}$  value increased to the maximum load of the loading cycle, the  $P_{\rm I}$  value hindered the slip-effect of the  $P_{\rm II}$ value parallel to the crack propagation direction and the  $K_{\rm II}$  value eventually stabilized as the crack continued to propagate, as shown at point c in Figure 6b. Due to the micro-geometric structure of the material and the influence of inclusions and defects in this process, the crack propagation path changed slightly, but the overall propagation path direction was approximately perpendicular to the loading axis.

Before analyzing the variation of SIFs during the crack deflection process, it is essential to understand how the applied load is projected into the directions parallel and perpendicular to the crack propagation under mixedmode conditions. This is crucial for gaining a clear insight into the changes in SIFs in Figure 6. With an increase in the initial loading angle, the  $P_{\rm I}$  value decreased, and the  $P_{\rm II}$  value increased. The variation in crack deflection angle leads to changes in the projected loads in different directions,  $P_{I}$  and  $P_{II}$  interact with each other in this process. The presence of  $P_{I}$  hinders the slip effect parallel to the crack propagation direction caused by  $P_{II}$ , while the presence of  $P_{\rm II}$  results in the deflection of the crack propagation direction.  $P_{\rm I}$  and  $P_{\rm II}$  dominated the variation trends of  $K_{I}$  and  $K_{II}$ , the final manifestation of the competitive relationship between  $P_{\rm I}$  and  $P_{\rm II}$  was the competitive relationship between  $K_{I}$ -dominated PICC and  $K_{II}$ -dominated RICC. The results indicated that the crack deflected in



Figure 6 Relationship between  $K_{l}$ ,  $K_{ll}$  and distance along crack propagation path with different loading angles (R=0.1): **a** a- $K_{l}$ , **b** a- $K_{ll}$ 



Figure 7 Loading force decomposition diagram with different loading angles: a  $a=30^\circ$ , b  $a=45^\circ$ , c  $a=60^\circ$ 

the first expansion stage and the crack propagation direction was perpendicular to the loading axis in the second.

The alternating effect of the loading angle  $\alpha$ =45° in Figure 6a was not apparent compared with other loading angles, and the reason for this result was analyzed. Under the initial condition of  $\alpha$ =45°, the values of  $P_{\rm I}$  and  $P_{\rm II}$  were equal, as shown in Figure 7b, and the crack deflected after a certain number of cycles. Although RICC has an effect on the PICC during the deflection process, the effect of the interaction was not significant due to the same magnitude of the load, so the peak value of  $K_{\rm II}$  at  $\alpha$ =45° in the first stage of Figure 6b was significantly lower than that at the other two angles. In addition, the mean value of  $K_{\rm II}$  value under three loading angles in the range of 2–5 mm was maintained at the same level, because the crack deflect needs to undergo a certain number of cycles.

## 4.2 Experimental Results

The roughness values ( $R_A$ ) on both sides of the crack in the fatigue crack propagation test were measured by the optical microscope (Atometrics ER230) after the test, and the crack roughness-induced closure scale was analyzed and calculated. In addition, the fracture morphology was analyzed by SEM (JSM 7800F). When measuring the roughness of the fracture surface, the fracture surface area is divided into four regions ((a)–(d)) according to the Paris crack propagation curve division method, which represents the prefabricated crack, low-speed, mediumspeed, and high-speed propagation areas, respectively, as shown in Figure 8a-d.

The crack surface  $R_A$  was measured, as shown in Figure 9a-d, the coordinate axes x and y represent the measurement range of roughness measurement. The measurement results in Figure 9a-d are derived from the corresponding positions in the region outlined in Figure 8a-d. The size of the measured fracture surface area is 900 × 500 µm. The result showed that the surface roughness on both sides of the crack gradually increased with crack propagation. The RICC value can be obtained by substituting the measurement results into Eq. (10). The reason for the change of roughness in Figure 9 is that the fatigue fracture surface contacts during the crack deflection process. Under the action of cyclic load, the friction of the contact surface leads to the increase of the roughness value of the fracture surface.

The calculation results of PICC and RICC obtained from the test are plotted in Figure 10. The overall level of the PICC value in Figure 10(a) was approximately 0.2, which was consistent with the theoretical calculating values of Eq. (8). However, the PICC value in the curve was not fixed at a certain value, and the PICC curve extended in a wavy manner. The final crack closure degree  $P_{op}/P_{max}$ 



**Figure 8** Fracture surfaces under mixed-mode (*R*=0.1, *a*=45°): **a** prefabricated crack stage, **b** initial crack deflection stage, **c** crack deflection stage, **d** end stage of crack deflection

tended to be constant at approximately 0.2, it was similar to the calculation results of Wang et al. [34] and Lu et al. [38]. The results from Wang et al. [34] also indicate an inconsistent trend in the  $P_{\rm op}/P_{\rm max}$  variation between the base material and the welded joint. This inconsistency is attributed to the fact that residual stresses in the weld are not eliminated through post-weld heat treatment. As the crack propagates, the  $P_{\rm op}/P_{\rm max}$  decreases and gradually stabilizes. Since the material used in this test was EA4T base metal, the crack closure trend of the material after welding was not studied.

The abscissa in Figure 10b represents the experimentally measured fracture surface  $R_A$ , while the ordinate represents the RICC results corresponding to different  $R_{\rm A}$ . The four measuring points in Figure 10b correspond to the pre-crack, the initial crack deflection, the middle crack deflection and the end of the crack deflection stage. A/B/C/D is used to represent the four stages for the convenience of analysis and description. The variation range of  $R_A$  from stage B to C was significantly narrower than the stage A to B in Figure 10b, it was the result of the interaction between the PICC and RICC. The interaction or competition between the two closure effects aggravates the contact between the crack fracture surfaces, resulting in a large change in the  $R_A$  value of the fracture surface. The RICC value presented in Figure 10b was illustrated by combining the simulation results shown in Figure 6b. The  $K_{\rm II}$  value provides the driving force for RICC, and the RICC effect was correspondingly enhanced with the increase of  $K_{II}$ . After the crack propagation entered stage D, the crack deflection ended and the crack propagation direction was approximately perpendicular to the loading axis. The change after stage D was not analyzed and discussed, as this study focuses on the crack deflection process under mixed-mode loading.

Figure 11 shows the recorded results of the fatigue crack growth rates (FCGR) during the experimental process: the FCGRs showed an overall upward trend, and there was also an alternating pattern of increase and decrease in the early stages of crack propagation. This trend was consistent with the results in Figure 6a, where  $K_{\rm I}$  values act as the driving force for crack propagation



Figure 9 Crack surface roughness measurement results (*R*=0.1, *a*=45°): **a** prefabricated crack stage, **b** initial crack deflection stage, **c** crack deflection stage, **d** end stage of crack deflection



Figure 10 Crack closure level with different loading angles: a PICC value, b RICC value

and PICC. The alternating changes in  $K_{\rm I}$  values contribute to the alternating pattern in FCGRs.

## 4.3 Discussions

Based on the above simulation and experiment, the authors surveyed relevant studies on crack closure under mixed-mode conditions. Partial experimental results were presented to conduct a comparative analysis with the findings of this study. Zhang et al's [52] simulation

results show that the crack closure level under the plane stress state is about 0.4, and the results are consistent with the estimated level of Eq. (8). The closure ratio in plane stress conditions is 2–2.5 times that in plane strain conditions, the calculated PICC average value of this test was 0.2, the experimental results in the plane strain state have the same trend as the above results. Furthermore, it was observed that as the stress ratio increased from 0.1 to 0.3, the degree of crack closure increased. This is



Figure 11 Test record result: a *a*-d*a*/d*N*, b  $\Delta K_{eff}$ -d*a*/d*N* 

attributed to the increase in  $P_{\rm min}$  in the denominator of Eq. (9). A higher stress ratio means that the loading minimum load  $P_{\rm min}$  increased, and the denominator in Eq. (9) decreased resulting in a significant upward shift of the crack closure level.

The FCG behaviors of Al-Li-Cu-Mg alloy (8090) and traditional Al-Cu-Mg alloy (2004) are compared in a study by Rodrigues et al. [39]. The crack closure level of Al-2024 is in the range of 0.4-0.6, but the crack closure level of Al-8090 alloy is 0.7–0.9, the reason is that the fracture analysis results of Al-8090 alloy indicated a transgranular fracture mode (high roughness, high crack deflection), it is characteristic of extensive crack closure caused by the roughness wedge. Suresh et al. [53] proposed the RICC geometric model of a fracture surface and considered the contribution of the crack tip displacement of modes I and II in early research. The results show that the displacement ratio is positively correlated with the crack closure level. The crack closure level corresponding to the test data of the two materials (Rail steel and 1018 steel) is approximately 0.4, which is of the same order of magnitude as the crack closure level of the PICC under CA in Eq. (8). The material used in this experiment was EA4T steel, which exhibits consistency in crack closure behavior with the material properties used in Suresh's experimental study.

The crack deflected in the prefabricated direction follows the same mechanism as the composite direction of the force defined in physics under mixed loading, as illustrated in Figure 7. As the deflection angle changes, the projected values onto the X and Y axes change accordingly. The plastic zone at the crack tip caused by angle deflection was a butterfly asymmetric structure. The asymmetric structure leads to crack propagation along the smaller side of the plastic zone. The crack propagation rate was slowed down because the load in the crack tip region generated the residual compressive stress. When the crack is deflected, the constraint of the crack along the propagation direction is reduced due to the smaller load projected onto mode-II. At this moment, the crack propagation along the single-slip system was dominated by mode-I load, resulting in a jagged crack propagation path. The contact area of the serrated crack propagation path increased as the load projected onto mode-II increased and the crack deflection progressed. The crack tip was passivated under cyclic loading, and the ratio of  $P_{\rm op}$  to  $P_{\rm max}$  approximately represented the PICC value. The presence of RICC caused the crack side to make contact in advance, resulting in a small  $P_{op}$  value in the subsequent loading cycle. The change in  $P_{op}$  was positively correlated with the PICC value, thus the PICC value fluctuates under cyclic loading. After crack deflection was completed, the crack propagation direction was approximately perpendicular to the loading axis. The crack propagation can be approximated as mode-I crack propagation, the PICC value tends to be a fixed value now, as shown in the latter half of the curve in Figure 10a. Moreover, the degree of crack closure was negatively correlated with the test loading angle, consistent with similar findings reported in Ref. [52].

As the contact area of the serrated crack propagation path increased, the contact friction between the fracture surfaces caused a larger  $R_A$  value in the contact area, as described in the calculation expression of the RICC value



Figure 12 Deflection diagram of plastic zone at crack tip ( $a=30^\circ$ , a=7.152mm): a test image obtained by DIC; b shape of plastic zone at crack tip



**Figure 13** Mixed-mode loading fracture surface (a=30°): **a** macro schematic diagram, **b** partial enlargement diagram

in Eq. (10). The RICC value shows a positive correlation with  $R_A$  value. The RICC values obtained from the test results were plotted in Figure 10b, demonstrating an approximately linear relationship with the  $R_A$  value, with the RICC value being around 0.15. The crack tip displacement field was obtained by the DIC technique, and then the plastic zone at the crack tip was calculated using the Tresca yield criterion, the position of the solution area is shown in Figure 12a. During the crack deflection process, the plastic zone at the crack tip exhibited an asymmetrical butterfly shape, as shown in Figure 12b.



**Figure 14** Changes in plastic zone and fracture surface profile during mixed-mode crack propagation

The larger side of the plastic zone at the crack tip facilitated the elastic recovery of the plastic inclusion when the load was at its minimum, which led to a decrease in the distance between the plastic wakes on both sides of the crack [54]. However, due to the serrated propagation path of the crack, the upper and lower fracture surfaces are contacted in advance during the elastic recovery stage hindering the PICC process.

The fracture surface of the specimen was analyzed using SEM at the end of the test, as shown in Figure 13. Figure 13 shows the macroscopic schematic of the mixed-mode fracture surface and the local amplification diagram of the crack deflection line position, respectively. In Figure 13a, the black arrow indicates the crack propagation direction, the fracture surface was rough and had no metallic lustre. The fatigue arc and crack deflection line angle were approximately 25°, and the crack deflection angle was approximately 30°. Notably, the roughness of the crack surface before and after crack deflection was different, the roughness of the crack surface before crack deflection was smooth, as shown in Figure 13b. The crack formed a symmetrical butterfly-shaped plastic zone at the crack tip under mode-I loading. Owing to the mixed-mode loading conditions, irreversible residual shear deformation was introduced at the crack tip, and a spatially torsional crack surface was formed during crack propagation.

During the crack propagation process, the shape of the plastic zone changed due to contact with the crack surface. The alterations in the plastic zone and fracture surface profile before and after crack deflection in the mixed-mode are illustrated in Figure 14, where (1) – (3) represents the prefabricated crack stage, the crack deflection stage, the stage with crack propagation direction perpendicular to the loading axis, respectively. Crack deflection occurred in the second stage, the plastic zone at the crack tip was asymmetric during the deflection process. The crack propagation line was used as the dividing line, and the area of the plastic zone at the crack tip in the M and the N regions changed alternately. The results of the interaction between PICC and RICC during crack deflection were again confirmed.

Considering the influence of fracture surface roughness on crack propagation rate, from the findings in this work, the evolution law of the fracture surface roughness is explored in the following work, and the parameters of the fracture surface roughness evolution are set onto the mixed-mode crack growth path, and the crack growth process can be obtained by combining the XFEM method. It is worth mentioning that the specimen is always in a tensile state under a high-stress ratio, and the crack closure effect will be affected by the stress ratio, which is also the influencing factor to be considered in future work.

# 5 Conclusions

(1) The driving force of the PICC and RICC is the  $K_{\rm I}$  and  $K_{\rm II}$ , the  $K_{\rm I}$  and  $K_{\rm II}$  values in the mixed-mode increased with crack propagation, and the latter then decreased to a stable value. The reason for this decrease is that the crack propagation direction gradually becomes perpendicular to the loading axis after crack deflection, the mixed-mode loading can be approximated as mode-I loading at this moment. The  $P_{\rm I}$  value hindered the slip effect of the

 $P_{\rm II}$  value parallel to the crack propagation direction during the process of crack deflection.

- (2) The interaction relationship between  $K_{\rm I}$  and  $K_{\rm II}$  values during crack deflection slowed the crack propagation rate to a certain extent. After crack deflection, the crack propagation rate increased linearly. The asymmetric butterfly-shaped plastic zone at the crack tip caused by crack deflection facilitated crack propagation along the smaller side of the plastic zone.
- (3) The value of the PICC result was fluctuation due to early contact with the crack surface caused by RICC. The serrated crack propagation path hindered the elastic recovery of the plastic inclusion at the crack tip when the load was at its minimum. The result of the interaction between PICC and RICC was that the variation in roughness caused by contact friction between the fracture surfaces was slowed.
- (4) The roughness difference of the crack surface before and after crack deflection was obvious. Owing to the mixed-mode loading, irreversible residual shear deformation was introduced at the crack tip, and a spatially torsional crack surface was formed during crack propagation. The changing trend of the crack tip plastic zone area was consistent with the changing trend of the fracture surface profile roughness, the plastic zone at the crack tip was asymmetric during the deflection process.

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#### **Author Contributions**

BY was in charge of the whole trial; SW wrote the manuscript; SW and SZ carried out the experimental analyses and the numerical simulations; JL and SX guided the experiments, BY and SZ revised the manuscript. All authors read and approved the final manuscript.

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#### Data availability

Date will be available from the corresponding author on reasonable request.

#### Declarations

#### **Competing Interests**

The authors declare no competing financial interests.

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