Perspective

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# Intelligent block copolymer self-assembly towards IoT hardware components

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#### **Supplementary Information for**

# Intelligent block copolymer self-assembly towards IoT hardware components

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## Supplementary Note 1: BCP self-assembly

Block copolymers (BCPs) are a type of macromolecule composed of chemically distinct polymer chains that are covalently end-linked. The chemical incompatibility of the different polymer blocks leads to phase separation. However, owing to the covalent bonds between the chain ends, the separation is limited to within the length scale of the polymeric molecules, leading to the formation of nanosized domain structures<sup>1–3</sup>. As BCP thin films show highly ordered nanostructures with tunable nanodomain periodicity, they have attracted interest as templates for nanoelectronics fabrication.

The most studied BCPs are linear AB diblock copolymers. The theoretically derived phase diagram (see the figure, panel a) illustrates that the phase behaviour is governed by the volume fraction of the A block ( $f_A$ ) and the segregation strength parameter ( $\chi N$ , where  $\chi$  is the Flory–Huggins interaction parameter for the AB monomer pair and N is the total degree of polymerization). Various equilibrium microdomain morphologies have been identified, including lamellae (L), bicontinuous gyroid (G), hexagonally packed cylinders (C), bodycentred cubic spheres (S) and close-packed spheres (S<sub>cp</sub>)<sup>4,5</sup>.

A key parameter of BCP self-assembly is the length scale or spacing (d) of its periodic nanodomain morphology, which is directly related to the characteristic feature size of the resulting nanopattern. BCP *d* spacing is generally determined by the polymer molecular chain

size and  $\chi$  value, following the scaling relationship  $d \approx \chi^{1/6} N^{\delta}$ , where  $\delta = 1/2$  in the weak segregation limit ( $\chi N \approx 10.5$ ) and  $\delta = 2/3$  in the strong segregation limit ( $\chi N \gg 10.5$ )<sup>5</sup>. For conventional BCPs with modest  $\chi$  values (weak segregation) and molecular weights, the ordered structures typically exhibit d spacings of  $10-50 \text{ nm}^6$ . To access nanopatterns with a broader range of feature sizes, BCPs with different chemistry and chain architectures have been explored (see the figure, panel b). High- $\chi$  BCPs have been exploited for sub-10-nm-scale ultrafine semiconductor patterning<sup>6</sup>. To fulfill the critical counterbalancing condition for microphase separation ( $\chi N > 10.5$ ), BCPs with a low N (and thus small feature size structure of <10 nm) should possess a high  $\chi$  value for the thermodynamic stabilization of the nanoscale structures<sup>7,8</sup>. At the other end of the feature size range (>100 nm), bottlebrush BCPs, composed of linear backbone and high-density side chain, with high molecular weights are being investigated<sup>9</sup> for the self-assembly of ordered nanodomains with d spacings exceeding 100 nm. The ordering kinetics (that defines the speed of the BCP self-assembly and it is thus an essential processing parameter for high throughput nanopatterning) of bottlebrush BCPs can be much faster than those of linear BCPs of similar molecular weights, because of their extended chain conformation and reduced chain entanglements<sup>10</sup>.

### Supplementary Note 2: Directed self-assembly

Although BCP thin films self-assemble into nanodomain structures under random thermal fluctuation, without an external driving force, the lateral ordering of the nanoscale domains is generally poor and characterized by dense structural defects, such as dislocations and disclinations<sup>11</sup>. Direct assembly (DSA) strategies have been implemented to create BCP nanopatterns with a high level of lateral ordering and minimal structural defects. The two most successful DSA approaches are epitaxial self-assembly<sup>12</sup> and graphoepitaxy<sup>13</sup> (see the figure, panel c). Epitaxial self-assembly uses chemically pre-patterned surfaces to direct the self-

assembly of BCP thin films and control the lateral ordering of the nanodomains<sup>12,14</sup>. By contrast, graphoepitaxy uses substrates with patterned topographical features<sup>13,15</sup>. Selective wetting of one BCP component to the trench side walls drives the lateral ordering of BCP nanodomains, leading to a well-aligned morphology along the trenches. Although BCP nanopatterns with highly ordered lateral ordering and minimal defects can be achieved by both DSA strategies, the ultimate level of defect control, including dislocation, bridging, and clustering, is a longstanding challenge, especially for the fabrication of delicate and complex semiconductor devices (such as CPU and memory devices).



**Supplementary Fig. 1**| Theoretical thermodynamic phase diagram of diblock copolymer selfassembly, tunable self-assembled nanopattern size governed by Flory-Huggins interaction parameter and molecular architecture, DSA strategies by means of epitaxial self-assembly and graphoepitaxy, and widely used BCPs and their characteristics. Panel **a** adapted with permission from ref. 4, American Chemical Society.

Table S1 | Representative types of block copolymer and their chemical structure, characteristics, and IoT device applications. PS-*b*-PMMA has relatively low  $\chi$  value, so it is easy to induce vertically aligned pattern without topcoat. Moreover, etching rate of PMMA is much faster than PS, enabling facile creation of PS mask pattern for pattern transfer process. In the case of PS-*b*-PEO, hydrophilic PEO block can be swelled by polar solvent and adsorb the ion precursor. Moreover, it forms micelle in the solution, enabling easy fabrication of mesoporous thick film. In the case of PS-*b*-P2VP and PS-*b*-P4VP, pyridinic nitrogen of P2VP and P4VP block can be easily functionalized by proton, resulting in easy metal precursor adsorption and crosslinking. PS-*b*-PDMS has high  $\chi$  value and inorganic component at PDMS, so it is easy to generate small pitch size pattern with high etching selectivity.

Polymer	PS- <i>b</i> -PMMA	PS- <i>b</i> -PEO	PS-b-P2VP & PS-b-P4VP		PS-b-PDMS
Chemical Structure				*	
Character istics	Easy processability Vertical alignment without top-coat Moderate etching selectivity	Versatile precursors decoration Easy to fabricate mesoporous thick films Solution self-assembly using micellization	Metal precursor adsorbability Easy charge modification Solution self-assembly using micellization		High etching selectivity Small pitch size

IoT device	Sensors Gas sensor <sup>16–18</sup> , biosensor <sup>19–22</sup> , IR Sensor <sup>23</sup> Energy harvesting Triboelectric nanogenerator <sup>24</sup> , therm oelectric nanogenerator <sup>25</sup> Metalens and antireflection coating <sup>2</sup> <sub>6,27</sub> Security systems <sup>28</sup>	<b>Sensors</b> Gas sensor <sup>29</sup> , biosensor <sup>30</sup> Interactive tactile interfaces <sup>31</sup>	Sensors Biosensor <sup>32</sup> Energy harvesting Thermoelectric nanogenerator <sup>33,34</sup> , osmotic energy conversion <sup>35,36</sup> User interface Environment-interactive displays <sup>37–41</sup> , anti-reflection coating <sup>42</sup> Security systems <sup>43</sup>	<b>Sensors</b> Gas sensor <sup>44</sup> <b>Energy harvesting</b> Triboelectric nanogener ator <sup>45</sup>
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